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# XAFS Study on the Specific Deoxidation Behavior of Iron Titanate Catalyst for the Selective Catalytic Reduction of NO<sub>x</sub> with NH<sub>3</sub>

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The environmentally friendly catalyst iron titanate (FeTiO<sub>x</sub>) was reported to be very active for the selective catalytic reduction of NO<sub>x</sub> with NH<sub>3</sub> (NH<sub>3</sub>-SCR), with high N<sub>2</sub> selectivity and H<sub>2</sub>O/SO<sub>2</sub> durability in the medium temperature range, and the specific microstructure of iron titanate crystallites as the active phase was determined. In consideration of the probable existence of a redox cycle between Fe<sup>3+</sup> and Fe<sup>2+</sup> species in the NH<sub>3</sub>-SCR reaction, the deoxidation behavior of the FeTiO<sub>x</sub> catalyst in an H<sub>2</sub> temperature-programmed reduction process was studied extensively by X-ray absorption near-edge structure (XANES) and extended X-ray absorption fine-structure (EXAFS) methods. Owing to the presence of an electronic inductive effect between Fe and Ti species in the unique edge-shared Fe<sup>3+</sup>–(O)<sub>2</sub>–Ti<sup>4+</sup> structure, the reducibility of Fe<sup>3+</sup> species in the FeTiO<sub>x</sub> catalyst was greatly enhanced compared with that in

pristine Fe<sub>2</sub>O<sub>3</sub>, leading to the higher oxidation ability of Fe species in FeTiO<sub>x</sub>. In the H<sub>2</sub> temperature-programmed reduction process, the well-dispersed Fe<sup>3+</sup> species in iron titanate crystallites could be totally converted into Fe<sup>2+</sup> in the form of ilmenite FeTiO<sub>3</sub> below 500 °C, whereas pristine Fe<sub>2</sub>O<sub>3</sub> could only be reduced to Fe<sub>3</sub>O<sub>4</sub> up to this temperature point. The typical NH<sub>3</sub>-SCR reaction is usually conducted below 500 °C, and the enhanced oxidation ability of Fe<sup>3+</sup> species in FeTiO<sub>x</sub> catalyst is responsible for its excellent catalytic NO<sub>x</sub> reduction performance at low temperatures. Based on XANES linear fitting and EXAFS curve-fitting results, the specific deoxidation process of the FeTiO<sub>x</sub> catalyst was proposed, which can provide useful information for the characterization of the microstructure and redox ability of active sites simultaneously in mixed oxide catalysts for certain catalytic reactions.

## Introduction

The greatly increased demand for the use of fossil fuels including coal and oil nowadays has led to the significant growth in emission of air pollutants including particulate matter, sulfur dioxide, and nitrogen oxides (NO<sub>x</sub>), especially in the developing countries with rapid increase in gross domestic product. Owing to the widely used dust-removal equipment and desulfurizing units, the particulate matter and SO<sub>2</sub> can be effectively controlled to a certain extent, whereas the ultrafine particles and NO<sub>x</sub> become the key targets for air-pollutant control in the next steps. NO<sub>x</sub>, including NO and NO<sub>2</sub> emitted from stationary and mobile sources, are major air pollutants resulting in acid rain, photochemical smog, respiratory disease,<sup>[1,2]</sup> and possibly even severe haze pollution, and need to be controlled effectively in accordance with more and more stringent regulations worldwide. Selective catalytic reduction of NO<sub>x</sub> with NH<sub>3</sub> (NH<sub>3</sub>-SCR) under oxygen-rich conditions is one of the most efficient technologies for the catalytic removal of NO<sub>x</sub>, and the earliest commercial catalyst system was WO<sub>3</sub>- or MoO<sub>3</sub>-promoted V<sub>2</sub>O<sub>5</sub>/TiO<sub>2</sub>, with high NH<sub>3</sub>-SCR performance and SO<sub>2</sub> durability in the medium or high temperature ranges.<sup>[3–6]</sup> Although this catalyst system has been utilized for several decades in industry, some

disadvantages still exist such as a narrow operation temperature window, low N<sub>2</sub> selectivity and poor stability at high temperatures, and, more importantly, the biological toxicity of the active V<sub>2</sub>O<sub>5</sub> phase, restricting its further application especially for the NO<sub>x</sub> reduction process for mobile sources such as diesel engines.<sup>[3,7,8]</sup> Therefore, more and more researchers now devote themselves to the development of new, highly efficient and environmentally friendly NH<sub>3</sub>-SCR catalysts with improved low-temperature deNO<sub>x</sub> efficiency, hydrothermal stability or resistance to co-existing pollutants, including zeolite catalysts promoted by metal cations (e.g., Fe-ZSM-5, Fe-BEA, Cu-ZSM-5, Cu-beta, Cu-CHA)<sup>[9–20]</sup> and single metal oxide, supported type, or mixed oxide catalysts (e.g., γ-Fe<sub>2</sub>O<sub>3</sub> nanorods, Fe<sub>2</sub>O<sub>3</sub>/WO<sub>3</sub>/ZrO<sub>2</sub>, Ti<sub>0.9</sub>Fe<sub>0.1</sub>O<sub>2–δ</sub>, Ce-P-O, CeTiO<sub>x</sub>, CeWO<sub>x</sub>, MnO<sub>x</sub>, (Fe<sub>3–x</sub>Mn<sub>x</sub>)<sub>1–δ</sub>O<sub>4</sub>).<sup>[21–29]</sup>

In our previous study, we reported a novel, environmentally benign iron titanate (FeTiO<sub>x</sub>) catalyst for NH<sub>3</sub>-SCR of NO<sub>x</sub> prepared by a facile coprecipitation method, achieving high deNO<sub>x</sub> efficiency, N<sub>2</sub> selectivity, and H<sub>2</sub>O/SO<sub>2</sub> durability in the medium temperature range.<sup>[30–32]</sup> The NH<sub>3</sub>-SCR mechanisms in the low- and high-temperature ranges were proposed according to in situ diffuse reflectance infrared Fourier transform spectroscopy and mass spectrometry to detect the surface-adsorbed species and gaseous components, respectively, in the SCR reaction,<sup>[33]</sup> most likely involving the redox cycle of Fe<sup>3+</sup> ↔ Fe<sup>2+</sup> for the activation of reactants including NH<sub>3</sub> and NO. Owing to the strong interaction between Fe and Ti species,

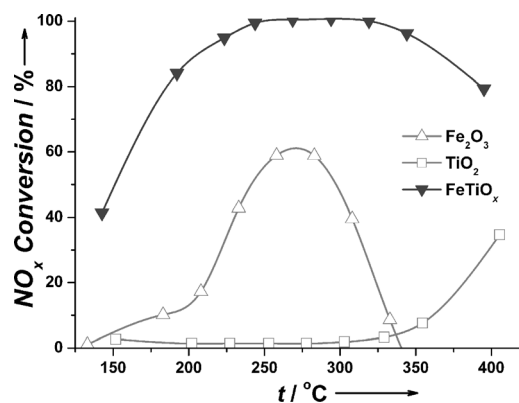
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the active phase in this  $\text{FeTiO}_x$  catalyst was mainly in the form of crystallites, the structure of which was rather difficult to characterize by conventional XRD, Raman spectra, or TEM, which usually require long-range order in the target materials. Therefore, the X-ray absorption fine structure (XAFS) method was used for the systematic structural characterization of this  $\text{FeTiO}_x$  catalyst,<sup>[34]</sup> which is very sensitive to the electronic state and local structure of specific central atoms. For the vanadium-free catalyst systems in the reported works mentioned above, the catalytic de $\text{NO}_x$  performance has usually been investigated in detail, yet the comprehensive investigation of their structure–activity relationships in the  $\text{NH}_3$ -SCR reaction is still largely lacking. In particular, many researchers have only paid attention to the microstructure of the active phase in  $\text{NH}_3$ -SCR catalysts and ignored the redox ability of the active species, which is actually not beneficial for the understanding of reaction mechanisms. Herein, the specific deoxidation behavior of Fe species in the  $\text{FeTiO}_x$  catalyst was sequentially investigated by using the XAFS method in combination with  $\text{H}_2$  temperature programmed reduction ( $\text{H}_2$ -TPR) experiments, which can provide useful information about the redox ability of Fe species having a special microstructure, to better understand the intrinsic reason for its high catalytic de $\text{NO}_x$  efficiency in the  $\text{NH}_3$ -SCR reaction.

## Results and Discussion

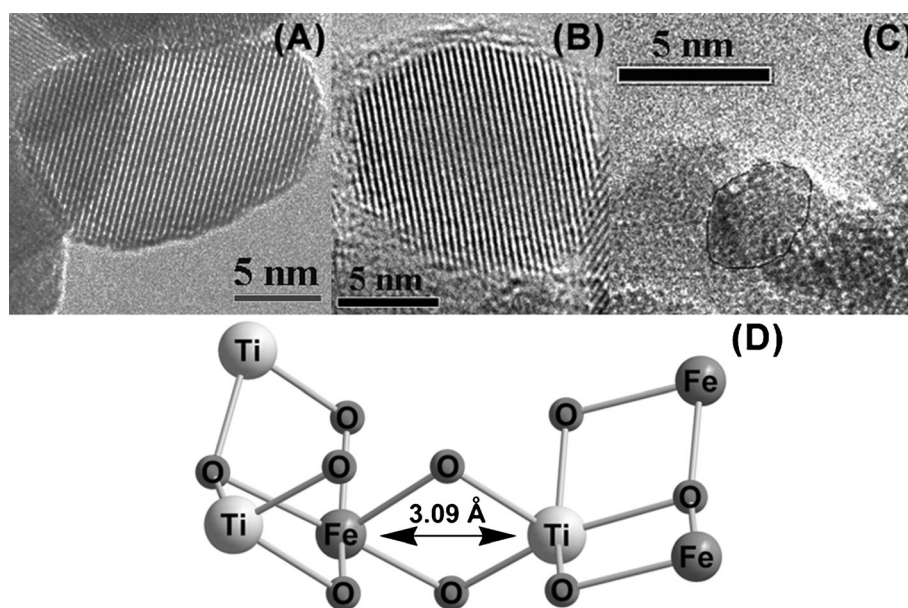
### $\text{NH}_3$ -SCR performance of the $\text{FeTiO}_x$ catalyst and its microstructure

The  $\text{NH}_3$ -SCR performance of the  $\text{FeTiO}_x$  catalyst compared with that of anatase  $\text{TiO}_2$  and hematite  $\text{Fe}_2\text{O}_3$  is shown in Figure 1. As we can clearly see, the  $\text{FeTiO}_x$  catalyst exhibited high  $\text{NH}_3$ -SCR performance over a broad temperature range, with over 80%  $\text{NO}_x$  conversion obtained from 200 to 400 °C. The pristine  $\text{TiO}_2$  prepared under our conditions exhibited no de $\text{NO}_x$  efficiency below 350 °C, and only 30%  $\text{NO}_x$  conversion could be achieved at temperatures as high as 400 °C. As for pristine  $\text{Fe}_2\text{O}_3$  also prepared by ourselves, the maximum  $\text{NO}_x$  conversion could only reach 60%, and the operation temperature window was rather narrow. These results indicate that the coexistence of Fe and Ti species in the  $\text{FeTiO}_x$  catalyst is important to obtain high de $\text{NO}_x$  efficiency, which can be attributed to the strong interaction between Fe and Ti species leading to larger surface areas, smaller particle sizes, higher NO oxidation ability and more abundant surface acid sites.<sup>[31,32]</sup>



**Figure 1.**  $\text{NO}_x$  conversions as a function of temperature in the  $\text{NH}_3$ -SCR reaction over  $\text{FeTiO}_x$  catalyst and references hematite  $\text{Fe}_2\text{O}_3$  and anatase  $\text{TiO}_2$ , respectively. Reaction conditions:  $[\text{NO}] = [\text{NH}_3] = 500$  ppm,  $[\text{O}_2] = 5$  vol%, 0.6 mL catalyst, and total flow rate of  $500 \text{ mL min}^{-1}$  yielding a gas hourly space velocity of  $50000 \text{ h}^{-1}$ .

As shown by the high-resolution TEM (HRTEM) images in Figure 2A and B,<sup>[32]</sup> both the anatase  $\text{TiO}_2$  ( $103.5 \text{ m}^2 \text{ g}^{-1}$ ) and hematite  $\text{Fe}_2\text{O}_3$  ( $42.5 \text{ m}^2 \text{ g}^{-1}$ ) were well crystallized with relatively large particle sizes, lacking rich surface defects for the adsorption and activation of reactants in the  $\text{NH}_3$ -SCR reaction. Possibly owing to the very similar ion radii of  $\text{Fe}^{3+}$  ( $0.645 \text{ \AA}$ ) and  $\text{Ti}^{4+}$  ( $0.605 \text{ \AA}$ ),<sup>[35]</sup> in the  $\text{FeTiO}_x$  catalyst the Fe species and Ti species could interdiffuse in a homogeneous state resulting in the formation of iron titanate crystallites with small particle size and relatively large surface area ( $245.3 \text{ m}^2 \text{ g}^{-1}$ ), which could supply more surface defects and active sites for the occurrence of the  $\text{NH}_3$ -SCR reaction. As indicated by the HRTEM image of the  $\text{FeTiO}_x$  catalyst (Figure 2C), no discernible lattice fringes could be observed, which was in good accordance with the broad diffraction bumps in XRD patterns and indistinct scattering bands in Raman spectra as presented in our previ-



**Figure 2.** HRTEM images of A) hematite  $\text{Fe}_2\text{O}_3$ , B) anatase  $\text{TiO}_2$ , and C)  $\text{FeTiO}_x$  catalyst,<sup>[32]</sup> and D) proposed structural model of  $\text{FeTiO}_x$  catalyst derived from EXAFS curve-fitting results<sup>[34]</sup>.

ous study.<sup>[31]</sup> Therefore, the XAFS method, which is very suitable to be used for the characterization of such amorphous or microcrystalline materials with no long-range order, was used to elucidate the microstructure of iron titanate crystallite in  $\text{FeTiO}_x$  catalyst in detail. Using X-ray absorption near-edge structure (XANES) and extended X-ray absorption fine structure (EXAFS) data in combination with DFT calculations to simulate the electronic density, we concluded that, different from the crystal structure of  $\text{Fe}^{3+}$  species in hematite  $\text{Fe}_2\text{O}_3$  and  $\text{Fe}_2\text{O}_3/\text{TiO}_2$  supported-type catalyst, a homogeneous  $\text{Fe}^{3+}-(\text{O})_2-\text{Ti}^{4+}$  structure was clearly formed in the  $\text{FeTiO}_x$  catalyst, in which the  $\text{Fe}^{3+}$  species and  $\text{Ti}^{4+}$  species were strongly linked in an edge-shared fashion (as shown by the structural model in Figure 2D).<sup>[34]</sup> The electronic inductive effect was confirmed to be present between  $\text{Fe}^{3+}$  species and  $\text{Ti}^{4+}$  species by XANES and X-ray photoelectron spectroscopy (XPS) results,<sup>[31,32,34]</sup> effectively reducing the electron density around  $\text{Fe}^{3+}$  species thus leading to the enhancement of NO adsorption, oxidation ability and finally the low temperature  $\text{NH}_3$ -SCR performance. Furthermore, the  $\text{Fe}^{3+}-(\text{O})_2-\text{Ti}^{4+}$  structure also showed high durability to  $\text{SO}_2$  poisoning in the  $\text{NH}_3$ -SCR reaction mainly attributable to the close and homogeneous combination of Fe species and Ti species as well as the easy decomposition of sulfate on Ti sites.<sup>[36]</sup> In previous studies by other researchers, the importance of the complex generated between the active phase and catalyst support was also emphasized, such as the Ag–O–Al species in Ag/ $\text{Al}_2\text{O}_3$  catalysts for the  $\text{CH}_4$ -SCR of  $\text{NO}_x$ <sup>[37]</sup> and the Ce–O–Ti species in  $\text{CeTiO}_x$  catalysts for the  $\text{NH}_3$ -SCR of  $\text{NO}_x$ .<sup>[38]</sup> Therefore, the study on the microstructure and chemical properties of such catalytically active species is an important and universal issue in the field of heterogeneous catalysis.

### $\text{H}_2$ -TPR results of $\text{FeTiO}_x$ catalyst and reference samples

Although the microstructure of iron titanate crystallites in the  $\text{FeTiO}_x$  catalyst has been clearly elucidated in our previous study, the reducibility of the Fe species has not been well recognized although it is actually very important for the completion of the redox cycle in the  $\text{NH}_3$ -SCR reaction. Therefore, an  $\text{H}_2$ -TPR experiment for the  $\text{FeTiO}_x$  catalyst was performed, using hematite  $\text{Fe}_2\text{O}_3$ , anatase  $\text{TiO}_2$ , and ilmenite  $\text{FeTiO}_3$  as references, and the results are shown in Figure 3.

As we can clearly see, no notable reduction peaks were present for pristine  $\text{TiO}_2$ , which indicates that the anatase support did not participate in the redox cycle for the  $\text{NH}_3$ -SCR reaction, owing to its lack of reducibility, only supplying adsorption sites for the reactants in the de $\text{NO}_x$  process. As for pristine  $\text{Fe}_2\text{O}_3$ , there were two  $\text{H}_2$  consumption peaks located at 405 and 725 °C, respectively. According to the baseline of the  $\text{H}_2$  signal, the Fe species in hematite  $\text{Fe}_2\text{O}_3$  started to be reduced at approximately 200 °C, similar to the initial reduction process of  $\gamma\text{-Fe}_2\text{O}_3$  nanorods prepared by Mou et al.<sup>[27]</sup> Possibly owing to the relatively large particle sizes and difficult  $\text{H}_2$  diffusion, the first reduction peak at 405 °C did not exhibit Gaussian symmetry, and the second reduction peak at 725 °C, overlapping with a shoulder reduction peak above 800 °C, did not return to the baseline until as high as 900 °C, which was different to a certain

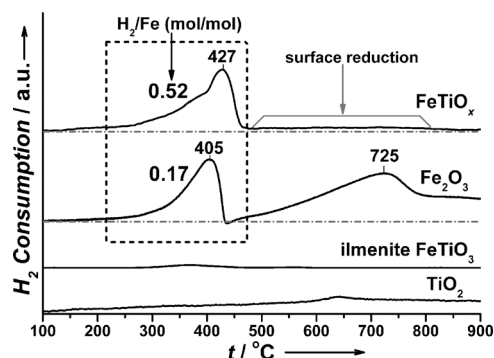


Figure 3.  $\text{H}_2$ -TPR profiles of  $\text{FeTiO}_x$  catalyst, hematite  $\text{Fe}_2\text{O}_3$ , anatase  $\text{TiO}_2$ ,<sup>[31]</sup> and ilmenite  $\text{FeTiO}_3$ .

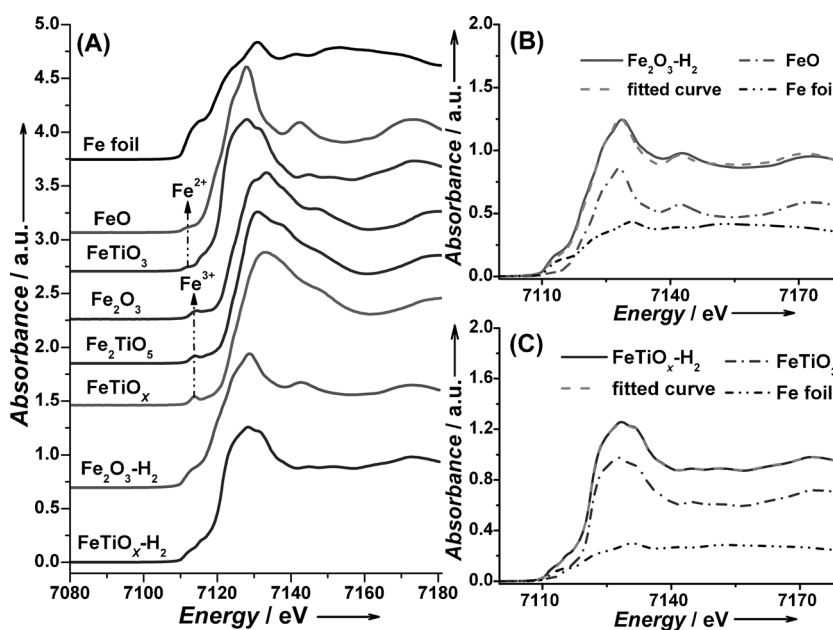
extent from the reduction processes observed for  $\gamma\text{-Fe}_2\text{O}_3$  nanorods enclosed by abundant reactive  $\{110\}$  and  $\{100\}$  crystal facets.<sup>[27]</sup> However, we could still calculate the  $\text{H}_2/\text{Fe}$  molar ratio for the reduction peak at low temperature to confirm the reduction process of Fe species in hematite  $\text{Fe}_2\text{O}_3$  below 500 °C, which is typically the maximum reaction temperature for the  $\text{NH}_3$ -SCR reaction. As shown in Figure 3, the  $\text{H}_2/\text{Fe}$  molar ratio for the reduction peak of  $\text{Fe}_2\text{O}_3$  at 405 °C was calculated to be 0.17, which was exactly consistent with the reduction process from  $\text{Fe}_2\text{O}_3$  to  $\text{Fe}_3\text{O}_4$ . Further raising the reduction temperature could lead to the formation of FeO or metallic  $\text{Fe}^0$ , yet different reduction mechanisms have been proposed by former researchers including the two-stage reduction of  $3\text{Fe}_2\text{O}_3 \rightarrow 2\text{Fe}_3\text{O}_4 + 6\text{Fe}$ <sup>[39,40]</sup> and the three-stage reduction of  $3\text{Fe}_2\text{O}_3 \rightarrow 2\text{Fe}_3\text{O}_4 + 6\text{FeO} \rightarrow 6\text{Fe}$ .<sup>[27,41,42]</sup> XANES measurements could clearly discriminate the possible mixed Fe species in reduced  $\text{Fe}_2\text{O}_3$ , which will be discussed in detail below.

Based on the  $\text{H}_2$ -TPR results of  $\text{TiO}_2$  and  $\text{Fe}_2\text{O}_3$ , the  $\text{H}_2$  reduction peaks observed for the  $\text{FeTiO}_x$  catalyst could be only ascribed to the reduction of Fe species, because the Ti species were barely reduced under these conditions. In the  $\text{FeTiO}_x$  catalyst with much higher  $\text{NH}_3$ -SCR performance than  $\text{TiO}_2$  and  $\text{Fe}_2\text{O}_3$ , the Fe species started to be reduced at approximately 150 °C, 50 °C lower than pristine  $\text{Fe}_2\text{O}_3$ . This result clearly shows that the oxygen species (mainly surface-adsorbed oxygen reduced at low temperatures) in the  $\text{FeTiO}_x$  catalyst is more labile probably owing to the strong interaction between Fe species and Ti species, leading to rich surface defects for the activation of oxygen. Interestingly, totally different from the reduction processes of  $\text{Fe}_2\text{O}_3$ , only one composite reduction peak was observed for the  $\text{FeTiO}_x$  catalyst with the maximum  $\text{H}_2$  consumption rate at 427 °C. According to the calculated  $\text{H}_2/\text{Fe}$  molar ratio of 0.52, the  $\text{Fe}^{3+}$  species in the  $\text{FeTiO}_x$  catalyst could be totally reduced to  $\text{Fe}^{2+}$  species below 500 °C, probably resulting in the transformation of iron titanate crystallite  $\text{Fe}^{3+}\text{TiO}_x$  to ilmenite  $\text{Fe}^{2+}\text{TiO}_3$ . Although the temperature point for the maximum  $\text{H}_2$  consumption rate of the  $\text{FeTiO}_x$  catalyst (427 °C) was higher than that of hematite  $\text{Fe}_2\text{O}_3$  (405 °C), the reducibility of Fe species in the former catalyst was actually much higher than that in the latter sample. Therefore, the synergistic combination of Fe species and Ti species in the  $\text{FeTiO}_x$  catalyst not only produced a unique edge-shared  $\text{Fe}^{3+}-(\text{O})_2-$

Ti<sup>4+</sup> structure, but also resulted in the easier deoxidation behavior of Fe species at low temperatures, which is beneficial to the completion of the redox cycle for the NH<sub>3</sub>-SCR reaction and thus the improvement of low-temperature deNO<sub>x</sub> efficiency. A similar phenomenon has also been reported by Li et al.,<sup>[38]</sup> in which the Ce species in the form of Ce–O–Ti linkages in a CeTiO<sub>x</sub> catalyst with excellent SCR performance showed higher reducibility than that in pristine CeO<sub>2</sub>. Above 500 °C, the H<sub>2</sub> reduction curve of the FeTiO<sub>x</sub> catalyst was slightly above the baseline, which could be attributed to the reduction of surface layers of Fe<sup>2+</sup>TiO<sub>3</sub> to metallic Fe<sup>0</sup> and leaving TiO<sub>2</sub> (probably in the rutile phase after 900 °C reduction). The in situ formed surface metallic Fe<sup>0</sup> might form clusters or particles on the Fe<sup>2+</sup>TiO<sub>3</sub> phase mainly caused by the relatively weak interaction between metal and oxide support. At the same time, no further reduction of Fe<sup>2+</sup> species in Fe<sup>2+</sup>TiO<sub>3</sub> phase was observed even at temperatures as high as 900 °C. A similar phenomenon was also found on the ilmenite FeTiO<sub>3</sub> reference, for which the H<sub>2</sub>-TPR profile revealed no obvious reduction peak of Fe<sup>2+</sup> species in the whole temperature range. This is another interesting characteristic of the H<sub>2</sub> reduction behavior of the FeTiO<sub>x</sub> catalyst, and the abovementioned points of view are verified by the following XANES linear-fitting results and EXAFS curve-fitting results.

### XANES results of Fe K, Ti K edges

Using Fe foil, FeO, FeTiO<sub>3</sub>, Fe<sub>2</sub>O<sub>3</sub>, and Fe<sub>2</sub>TiO<sub>5</sub> as references, the XANES spectra of the FeK edge in FeTiO<sub>x</sub> catalysts before and after H<sub>2</sub> reduction together with H<sub>2</sub>-reduced Fe<sub>2</sub>O<sub>3</sub> were recorded, as shown in Figure 4A. According to the pre-edge peak position, the Fe species in the FeTiO<sub>x</sub> catalyst was mainly present in the form of Fe<sup>3+</sup>, similar to those in Fe<sub>2</sub>O<sub>3</sub> and Fe<sub>2</sub>TiO<sub>5</sub>. According to the first-order derivatives of FeK XANES as reported in our previous study,<sup>[34]</sup> the absorption edge energy of Fe<sup>3+</sup> species in the FeTiO<sub>x</sub> catalyst (7123.6 eV) was higher than those in Fe<sub>2</sub>O<sub>3</sub> (7123.2 eV) and Fe<sub>2</sub>TiO<sub>5</sub> (7123.4 eV), mainly owing to the presence of the most abundant Fe<sup>3+</sup>–(O)<sub>2</sub>–Ti<sup>4+</sup> structure with an electronic inductive effect between Fe<sup>3+</sup> and Ti<sup>4+</sup> species, resulting in the higher oxidation ability of Fe<sup>3+</sup> species. This result confirmed by XPS data in our previous study<sup>[31]</sup> is also in good accordance with the H<sub>2</sub>-TPR results, in which the Fe<sup>3+</sup> species in the FeTiO<sub>x</sub> catalyst is much more easily reduced than that in pristine Fe<sub>2</sub>O<sub>3</sub>. Besides, the highest pre-edge peak intensity and the smoothest post-edge region



**Figure 4.** A) Normalized XANES spectra of FeK edge in Fe-containing samples; B) XANES linear-fitting results of FeK edge in hematite Fe<sub>2</sub>O<sub>3</sub> after H<sub>2</sub> reduction using FeO and Fe as references; C) XANES linear-fitting results of FeK edge in FeTiO<sub>x</sub> catalyst after H<sub>2</sub> reduction using FeTiO<sub>3</sub> and Fe as references. Samples specified by “–H<sub>2</sub>” refer to samples after the H<sub>2</sub>-TPR procedure to 900 °C; see Experimental Section.

of FeK XANES in the FeTiO<sub>x</sub> catalyst indicate that the Fe<sup>3+</sup> species in this catalyst exhibits the most severe structure distortion and the smallest particle size, which is beneficial to the generation of rich surface defects and thus high catalytic performance. The similarity of FeK XANES patterns between the FeTiO<sub>x</sub> catalyst and the Fe<sub>2</sub>TiO<sub>5</sub> reference indicates that the iron titanate crystallites may have some Fe–O or Fe–O–Ti coordination shells similar to those in Fe<sub>2</sub>TiO<sub>5</sub>, which can be verified by EXAFS results. As for the H<sub>2</sub>-reduced Fe<sub>2</sub>O<sub>3</sub> and FeTiO<sub>x</sub> samples, both XANES patterns of the FeK edge showed marked changes, with the shift of absorption edges towards the lower energy range and the substitution of the former pre-edge peaks by new pre-edge peaks characteristic of metallic Fe<sup>0</sup>, mixing with other pre-edge peak features. The post-edge regions of FeK XANES in H<sub>2</sub>-reduced Fe<sub>2</sub>O<sub>3</sub> and FeTiO<sub>x</sub> samples also exhibited mixed patterns, not similar to those of either FeO or FeTiO<sub>3</sub> individually. Therefore, FeK XANES linear fitting was performed for Fe<sub>2</sub>O<sub>3</sub> and FeTiO<sub>x</sub> samples after H<sub>2</sub> reduction by using FeO, FeTiO<sub>3</sub> and Fe foil as references to better confirm the existing state of Fe species in these H<sub>2</sub>-reduced samples, and the fitting results are shown in Figure 4B and 4C, respectively. By linear combination of FeK XANES data in FeO and Fe foil, the XANES spectrum of H<sub>2</sub>-reduced Fe<sub>2</sub>O<sub>3</sub> is well simulated with high fitting degree. As the fitting data in Table 1 reveal, the contribution of FeO and metallic Fe to the overall FeK XANES of H<sub>2</sub>-reduced Fe<sub>2</sub>O<sub>3</sub> is approximately 66.3% and 33.7%, respectively. This result suggests that after the reduction of Fe<sub>2</sub>O<sub>3</sub> to Fe<sub>3</sub>O<sub>4</sub> below 500 °C, the Fe species could be reduced further to FeO and then to metallic Fe<sup>0</sup> mainly following a three-stage reduction mechanism as mentioned above.<sup>[27,41,42]</sup> As for the H<sub>2</sub>-reduced FeTiO<sub>x</sub> catalyst, no FeO phase can be abstracted from the FeK XANES spectrum, con-



**Table 1.** XANES linear-fitting results of FeK edge and TiK edge in hematite  $\text{Fe}_2\text{O}_3$  and  $\text{FeTiO}_x$  catalyst after  $\text{H}_2$  reduction.

Sample	FeK XANES		TiK XANES	
	Ref.	Molar ratio [%]	Ref.	Molar ratio [%]
$\text{Fe}_2\text{O}_3\text{-H}_2$	FeO	66.3	–	–
	Fe foil	33.7	–	–
$\text{FeTiO}_x\text{-H}_2$	$\text{FeTiO}_3$	76.6	$\text{FeTiO}_3$	72.5
	Fe foil	23.4	rutile $\text{TiO}_2$	27.5

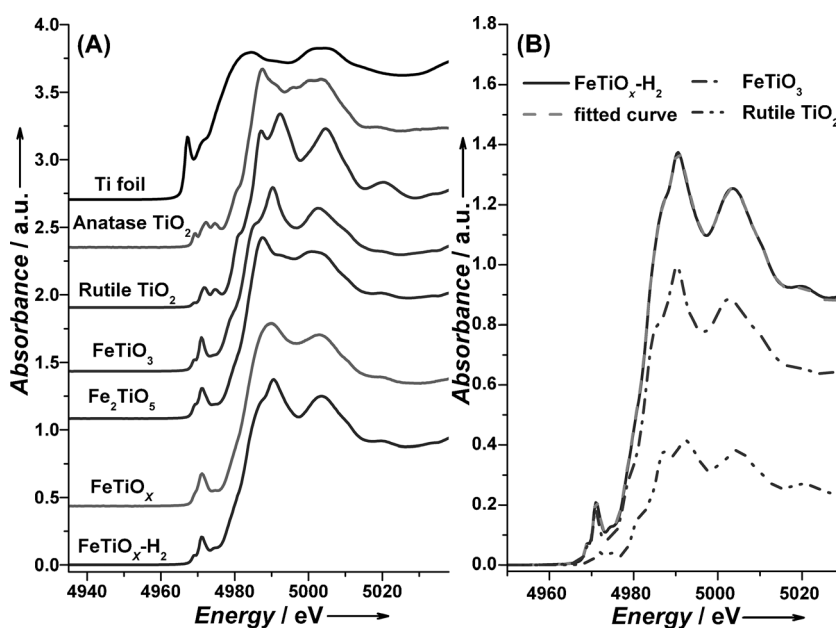
firming again that no  $\text{Fe}_2\text{O}_3$  phase existed in the as-prepared  $\text{FeTiO}_x$  catalyst and a new structure of Fe species was indeed formed. Using  $\text{FeTiO}_3$  and Fe foil as references, the FeK XANES of the  $\text{FeTiO}_x$  catalyst after  $\text{H}_2$  reduction is well simulated with high fitting degree, which is consistent with our viewpoint concerning the  $\text{H}_2$ -TPR result in Figure 3. As shown in Table 1, the contribution of  $\text{FeTiO}_3$  and metallic Fe to the overall FeK XANES of the  $\text{FeTiO}_x$  catalyst after  $\text{H}_2$  reduction is approximately 76.6% and 23.4%, respectively. Owing to the homogeneous state of the  $\text{Fe}^{3+}\text{-(O)}_2\text{-Ti}^{4+}$  structure in the iron titanate crystallites, the  $\text{FeTiO}_x$  catalyst could be uniformly transformed into  $\text{FeTiO}_3$  at relatively low temperatures, revealing a deoxidation behavior totally different from that of pristine  $\text{Fe}_2\text{O}_3$ . Further increasing the reduction temperature indeed only led to the formation of metallic  $\text{Fe}^0$  cluster or particles, leaving a majority of Fe species in the form of ilmenite  $\text{FeTiO}_3$ . Owing to the stoichiometric balance between Fe and Ti species (1:1) in the as-prepared  $\text{FeTiO}_x$  catalyst, the formation of metallic  $\text{Fe}^0$  in the  $\text{H}_2$ -reduced sample must result in the formation of excess  $\text{TiO}_2$  phase, which can be verified by the following TiK XANES results.

The XANES spectra of TiK edge in  $\text{FeTiO}_x$  catalysts before and after  $\text{H}_2$  reduction were recorded by using Ti foil, anatase  $\text{TiO}_2$ , rutile  $\text{TiO}_2$ ,  $\text{FeTiO}_3$ , and  $\text{Fe}_2\text{TiO}_5$  as references, and the results are shown in Figure 5A. As we can clearly see, not only the pre-edge peak shape but also the post-edge region of TiK XANES in the as-prepared  $\text{FeTiO}_x$  catalyst was totally different from those in anatase  $\text{TiO}_2$  and rutile  $\text{TiO}_2$ , indicating the presence of  $\text{Ti}^{4+}$  species in another coordination state. Notably, the XANES pattern of the TiK edge in  $\text{FeTiO}_x$  catalyst was similar to that in  $\text{Fe}_2\text{TiO}_5$  but with higher pre-edge peak intensity and smoother post-edge region, suggesting that the iron titanate crystallites with smaller particle sizes have Ti–O or Ti–O–Fe coordination shells similar to those in  $\text{Fe}_2\text{TiO}_5$  yet with more severe structure distortion, similar to

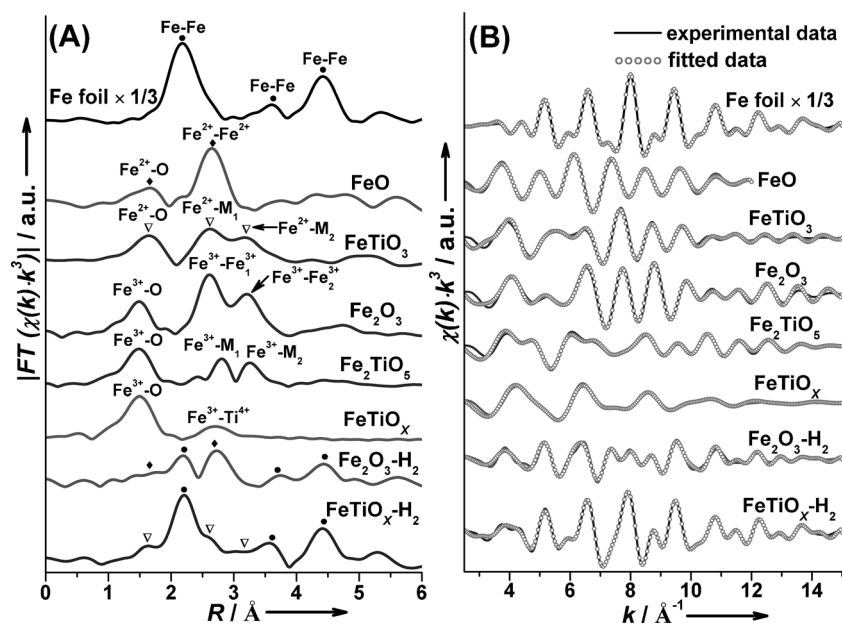
the conclusions from the FeK XANES experiments. Based on the first-order derivatives of TiK XANES in our previous study,<sup>[34]</sup> the absorption edge energy of  $\text{Ti}^{4+}$  species in the  $\text{FeTiO}_x$  catalyst (4983.4 eV) was lower than those in anatase  $\text{TiO}_2$  (4985.5 eV) and  $\text{Fe}_2\text{TiO}_5$  (4984.3 eV), which can be simply explained by the reverse electronic inductive effect between  $\text{Fe}^{3+}$  and  $\text{Ti}^{4+}$  species in the most abundant  $\text{Fe}^{3+}\text{-(O)}_2\text{-Ti}^{4+}$  structures. As for the  $\text{H}_2$ -reduced  $\text{FeTiO}_x$  catalyst, both the pre-edge peak and post-edge region revealed a marked change, exhibiting a mixed state of  $\text{Ti}^{4+}$  species. As presented in Figure 5B, the XANES spectrum of the TiK edge in the  $\text{FeTiO}_x$  catalyst after  $\text{H}_2$  reduction can be well simulated by using ilmenite  $\text{FeTiO}_3$  and rutile  $\text{TiO}_2$  as references, which is in good accordance with the  $\text{H}_2$ -TPR and FeK XANES results. The contributions of ilmenite  $\text{FeTiO}_3$  and rutile  $\text{TiO}_2$  to the overall XANES result from the perspective of TiK edge are approximately 72.5% and 27.5%, respectively (Table 1), confirming again that the formation of metallic  $\text{Fe}^0$  resulted in the presence of residual rutile  $\text{TiO}_2$  in an approximately equivalent amount. In brief, not only from the FeK XANES results but also from the TiK XANES results, we can conclude that the  $\text{Fe}^{3+}$  and  $\text{Ti}^{4+}$  species are homogeneously dispersed at the atomic scale, and the electronic inductive effect between them results in the enhancement of oxidation ability of Fe species leading to the more facile deoxidation behavior of the  $\text{FeTiO}_x$  catalyst, which is beneficial to the completion of the redox cycle in the  $\text{NH}_3$ -SCR reaction and thus the promotion of  $\text{deNO}_x$  efficiency.

#### EXAFS results of FeK and TiK edges

To further confirm the existing states of Fe species in the  $\text{FeTiO}_x$  catalyst and  $\text{Fe}_2\text{O}_3$  sample after  $\text{H}_2$  reduction, the EXAFS data of the FeK edge were curve-fitted by using the least-square method. In Figure 6A, the Fourier transforms of filtered



**Figure 5.** A) Normalized XANES spectra of TiK edge in Ti-containing samples; B) XANES linear fitting results of TiK edge in  $\text{FeTiO}_x$  catalyst after  $\text{H}_2$  reduction using  $\text{FeTiO}_3$  and Fe as references. Samples specified by “ $\text{-H}_2$ ” refer to samples after the  $\text{H}_2$ -TPR procedure to 900 °C; see Experimental Section.



**Figure 6.** A) Fourier transforms of filtered EXAFS oscillations  $k^3\chi(k)$  into  $R$  space and B) corresponding filtered  $k^3\chi(k)$  of FeK edge in Fe-containing samples in the  $k$  range of 2.5–15  $\text{\AA}^{-1}$ . The dotted lines correspond to the curve-fitting results in  $k$  space calculated by using the FEFF 8.4 code. ● Indicate  $\text{Fe}^0$ ; ◆  $\text{FeO}$ ; ▽  $\text{FeTiO}_3$ .

EXAFS oscillations  $k^3\chi(k)$  of the FeK edge in all Fe-containing samples into  $R$  space in the  $k$  range of 2.5–15  $\text{\AA}^{-1}$  are shown, and in Figure 6B, the corresponding filtered  $k^3\chi(k)$  plots are shown, in which the dotted lines correspond to the curve-fitting results in  $k$  space calculated by using the FEFF 8.4 code with different reference models. The corresponding curve-fitted data of the FeK edge are shown in Table 2. The as-prepared  $\text{FeTiO}_x$  catalyst clearly exhibited a well-defined first Fe–O coordination shell and a rather small second coordination shell, in complete contrast to  $\text{Fe}_2\text{O}_3$  or  $\text{Fe}_2\text{TiO}_5$ , which exhibit clear bimodal coordination peaks. In our previous study,<sup>[34]</sup> this unique second coordination shell was attributed to the scattering pathway of Fe–O–Ti, and the Fe–O–Ti bond length was calculated to be 3.09  $\text{\AA}$ . This bond length was shorter than that in well-crystallized  $\text{Fe}_2\text{TiO}_5$  (3.18  $\text{\AA}$ ) and belonged to the edge-shared  $\text{Fe}^{3+}-(\text{O})_2-\text{Ti}^{4+}$  structure, causing the electronic inductive effect between  $\text{Fe}^{3+}$  and  $\text{Ti}^{4+}$  species to take place more easily, thus resulting in its specific deoxidation behavior and high  $\text{NH}_3$ -SCR performance. The small coordination number of Fe–O–Ti (2.6) in the  $\text{FeTiO}_x$  catalyst is in good accordance with the XANES results, confirming again the existence of abundant structural defects in iron titanate crystallites, which is advantageous to the adsorption and activation of reactants for the de $\text{NO}_x$  process. After  $\text{H}_2$  reduction, both the  $\text{Fe}_2\text{O}_3$  sample and  $\text{FeTiO}_x$  catalyst presented coordination shells with a mixed state, for which the common peaks in Figure 6A (indicated by ●) can be attributed to metallic  $\text{Fe}^0$  as mentioned above. For  $\text{H}_2$ -reduced  $\text{Fe}_2\text{O}_3$ , the Fe–O and Fe–O–Fe shells belonging to FeO can be clearly observed (indicated by ◆). As for the  $\text{H}_2$ -reduced  $\text{FeTiO}_x$  catalyst, the Fe–O and Fe–O–M (M = Fe or Ti) shells ascribed to  $\text{FeTiO}_3$  can be observed (indicated by ▽), confirming again that the iron titanate crystallites can

be uniformly reduced to ilmenite  $\text{FeTiO}_3$  phase as a result of the homogeneous  $\text{Fe}^{3+}-(\text{O})_2-\text{Ti}^{4+}$  structure with unique deoxidation behavior. These conclusions are well supported by the curve-fitting results of FeK edge EXAFS (Table 2) with high degree of fit between the experimental data and FEFF-calculated data (Figure 6B). The relatively small coordination numbers of Fe–Fe shells in the  $\text{H}_2$ -reduced  $\text{FeTiO}_x$  catalyst also suggest that the metallic  $\text{Fe}^0$  formed after high-temperature reduction exhibited rather small particle sizes, probably resulting from the weak interaction between  $\text{Fe}^0$  and the oxide support and also the hindering of  $\text{Fe}^0$  aggregation by the in situ released  $\text{TiO}_2$ .

In a similar way, the TiK edge EXAFS data of  $\text{FeTiO}_x$  catalysts

before and after  $\text{H}_2$  reduction were curve-fitted by the least-square method using different Ti-containing reference models, with the Fourier transformed  $k^3\chi(k)$  shown in Figure 7A in the  $k$  range of 2.5–13  $\text{\AA}^{-1}$  and the filtered  $k^3\chi(k)$  shown in Figure 7B, in which the dotted lines correspond to the FEFF-calculated scattering pathways. The relevant curve-fitted data of TiK edge EXAFS results are shown in Table 3. Similar to the case of FeK edge EXAFS experiments, a well-defined first Ti–O coordination shell and a rather small second coordination shell can be observed for the  $\text{FeTiO}_x$  catalyst, the second shell of which is different from those in anatase  $\text{TiO}_2$ , rutile  $\text{TiO}_2$ , or  $\text{Fe}_2\text{TiO}_5$ , which reveal clear bimodal coordination peaks. According to the curve-fitting results, the small second coordination shell of the  $\text{FeTiO}_x$  catalyst can be attributed to the scattering pathway of Ti–O–Fe with smaller bond length (3.08  $\text{\AA}$ ) than that in  $\text{Fe}_2\text{TiO}_5$  (3.17  $\text{\AA}$ ). From the perspective of the central Ti species, this bond length can be ascribed to the edge-shared  $\text{Ti}^{4+}-(\text{O})_2-\text{Fe}^{3+}$  structure, in which the Ti–O–Fe coordination number is only 2.0 because of the crystallite phase of the  $\text{FeTiO}_x$  catalyst. After  $\text{H}_2$  reduction, not only the first but also the second coordination shells of Ti species revealed some notable variations, exhibiting overlapping peak shapes. Using ilmenite  $\text{FeTiO}_3$  and rutile  $\text{TiO}_2$  as references, the TiK EXAFS data in the  $\text{H}_2$ -reduced  $\text{FeTiO}_x$  catalyst could be well simulated with high fitting degree, which is quite consistent with the TiK edge XANES results.

### Proposed deoxidation process of the $\text{FeTiO}_x$ catalyst

Based on the above-mentioned  $\text{H}_2$ -TPR, XANES linear fitting, and EXAFS curve-fitting results, the specific deoxidation behavior of  $\text{FeTiO}_x$  catalyst can be proposed as shown in Scheme 1.

**Table 2.** Calibrated EXAFS curve-fitting results of FeK edge in different samples (M = Fe or Ti).

Sample	Ref. <sup>[a]</sup>	Shell	CN <sup>[b]</sup>	R <sup>[c]</sup> [Å]	ΔE [eV]	DW <sup>[d]</sup> [Å]	R factor [%]
Fe foil	Fe	Fe–Fe	8.0 ± 3.7	2.48 ± 0.03	–4.7 ± 7.4	0.072 ± 0.033	0.5
		Fe–Fe <sub>1</sub>	6.0 ± 6.8	2.87 ± 0.07	–7.1 ± 12.7	0.081 ± 0.080	
		Fe–Fe <sub>2</sub>	12.0 ± 9.6	4.05 ± 0.06	–11.3 ± 6.1	0.099 ± 0.063	
		Fe–Fe <sub>3</sub>	24.0 ± 7.7	4.75 ± 0.02	–4.1 ± 2.0	0.092 ± 0.024	
FeO	FeO	Fe–O	6.0 ± 2.3	2.10 ± 0.03	–11.2 ± 4.7	0.092 ± 0.051	1.8
		Fe–O–Fe	12.0 ± 2.6	3.06 ± 0.01	–8.0 ± 1.8	0.108 ± 0.018	
FeTiO <sub>3</sub>	FeTiO <sub>3</sub>	Fe–O	6.0 ± 1.2	2.10 ± 0.03	–7.9 ± 3.5	0.097 ± 0.030	6.1
		Fe–O–M <sub>1</sub>	4.0 ± 1.0	2.98 ± 0.02	2.1 ± 2.3	0.111 ± 0.022	
		Fe–O–M <sub>2</sub>	3.0 ± 1.0	3.38 ± 0.02	17.5 ± 2.5	0.074 ± 0.030	
Fe <sub>2</sub> O <sub>3</sub>	Fe <sub>2</sub> O <sub>3</sub>	Fe–O	6.0 ± 1.5	1.95 ± 0.03	–15.1 ± 5.4	0.101 ± 0.037	4.2
		Fe–O–Fe <sub>1</sub>	3.0 ± 0.6	2.97 ± 0.02	–2.8 ± 2.3	0.095 ± 0.017	
		Fe–O–Fe <sub>2</sub>	3.0 ± 1.0	3.36 ± 0.02	14.0 ± 3.2	0.064 ± 0.028	
Fe <sub>2</sub> TiO <sub>5</sub>	Fe <sub>2</sub> TiO <sub>5</sub>	Fe–O	3.0 ± 1.1	1.95 ± 0.02	–15.9 ± 3.3	0.089 ± 0.018	9.0
		Fe–O–M <sub>1</sub>	3.0 ± 2.3	3.18 ± 0.04	1.9 ± 7.1	0.072 ± 0.043	
		Fe–O–M <sub>2</sub>	3.0 ± 2.7	3.67 ± 0.05	–16.2 ± 7.1	0.093 ± 0.051	
FeTiO <sub>x</sub>	Fe <sub>2</sub> TiO <sub>5</sub>	Fe–O	7.7 ± 1.7	1.96 ± 0.01	–8.6 ± 2.2	0.095 ± 0.016	4.5
		Fe–O–Ti	2.6 ± 2.5	3.09 ± 0.04	4.6 ± 4.4	0.104 ± 0.045	
Fe <sub>2</sub> O <sub>3</sub> -H <sub>2</sub>	FeO *Fe	Fe–O	4.2 ± 2.9	2.10 ± 0.03	–5.2 ± 3.8	0.099 ± 0.071	7.1
		Fe–O–Fe	8.2 ± 1.8	3.06 ± 0.01	–4.0 ± 1.2	0.107 ± 0.01	
		*Fe–Fe	0.8 ± 0.4	2.48 ± 0.03	–2.5 ± 0.6	0.063 ± 0.029	
		*Fe–Fe <sub>1</sub>	0.5 ± 0.8	2.87 ± 0.07	–6.8 ± 12.2	0.067 ± 0.066	
		*Fe–Fe <sub>2</sub>	1.0 ± 1.6	4.05 ± 0.06	–7.9 ± 0.0	0.072 ± 0.046	
		*Fe–Fe <sub>3</sub>	2.2 ± 0.7	4.75 ± 0.02	–4.2 ± 11.3	0.075 ± 0.020	
FeTiO <sub>x</sub> -H <sub>2</sub>	FeTiO <sub>3</sub> *Fe	Fe–O	3.4 ± 2.5	2.10 ± 0.03	–4.4 ± 4.7	0.080 ± 0.081	1.2
		Fe–O–M <sub>1</sub>	2.3 ± 0.6	2.98 ± 0.02	1.8 ± 2.8	0.105 ± 0.021	
		Fe–O–M <sub>2</sub>	1.7 ± 2.9	3.38 ± 0.02	16.0 ± 0.0	0.070 ± 0.028	
		*Fe–Fe	1.5 ± 3.5	2.48 ± 0.03	–4.2 ± 0.1	0.058 ± 0.027	
		*Fe–Fe <sub>1</sub>	0.9 ± 2.2	2.87 ± 0.07	–12.6 ± 0.7	0.062 ± 0.061	
		*Fe–Fe <sub>2</sub>	2.3 ± 1.9	4.05 ± 0.06	–13.2 ± 1.1	0.077 ± 0.049	
		*Fe–Fe <sub>3</sub>	4.7 ± 2.9	4.75 ± 0.02	–3.7 ± 0.1	0.080 ± 0.021	

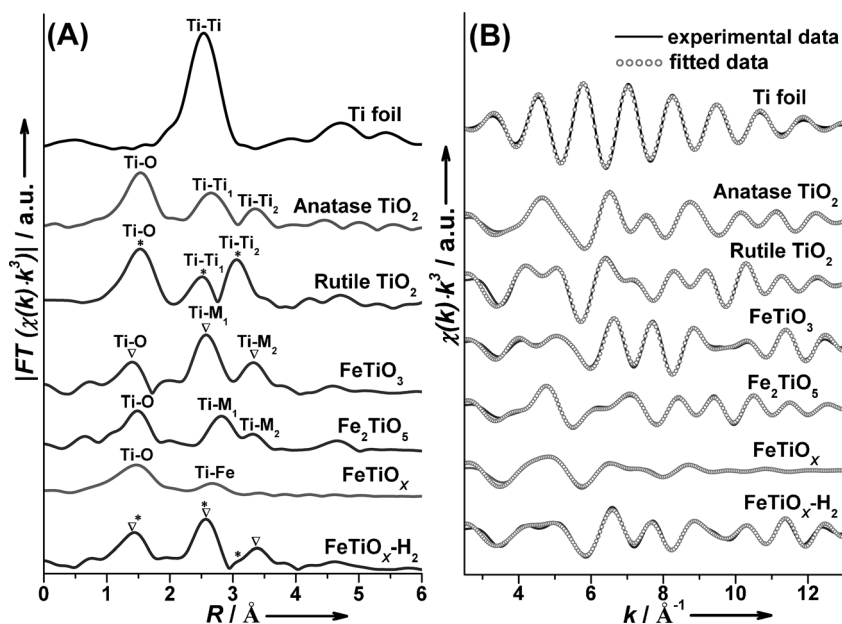
[a] For information on the calibration of (reference) coordination numbers and the treatment of error margins, see the Experimental Section. [b] Coordination number. [c] Bond length. [d] Debye–Waller factor.

Owing to the homogeneous state of Fe<sup>3+</sup> species and Ti<sup>4+</sup> species with electronic inductive effect in the edge-shared Fe<sup>3+</sup>–(O)<sub>2</sub>–Ti<sup>4+</sup> structure of iron titanate crystallites, the Fe<sup>3+</sup> species with higher oxidation ability can be totally converted into Fe<sup>2+</sup> species below 500 °C, which is a much easier process than the corresponding reduction of Fe<sup>3+</sup> species in pristine Fe<sub>2</sub>O<sub>3</sub>. It is noteworthy that the NH<sub>3</sub>-SCR reaction mainly occurs over the FeTiO<sub>x</sub> catalyst below 500 °C with high deNO<sub>x</sub> efficiency, which is closely related to the enhanced redox ability of Fe<sup>3+</sup> species in the specific Fe<sup>3+</sup>–(O)<sub>2</sub>–Ti<sup>4+</sup> structure. The as-formed Fe<sup>2+</sup> species in the H<sub>2</sub>-reduced sample combines with Ti<sup>4+</sup> species to exist as ilmenite FeTiO<sub>3</sub>, but not as FeO as in the Fe<sub>2</sub>O<sub>3</sub> case, indicating again that the Fe species and Ti species are present in a high dispersion state at the atomic scale. Above 500 °C, the surface layers of as-formed FeTiO<sub>3</sub> can be further reduced, which results in the formation of partial metallic Fe<sup>0</sup> clusters or particles and residual rutile TiO<sub>2</sub>. However, the reduction of Fe<sup>2+</sup> species in bulk FeTiO<sub>3</sub> phase cannot be achieved even at temperatures as high as 900 °C. In short summary, the unique microstructure of iron titanate crys-

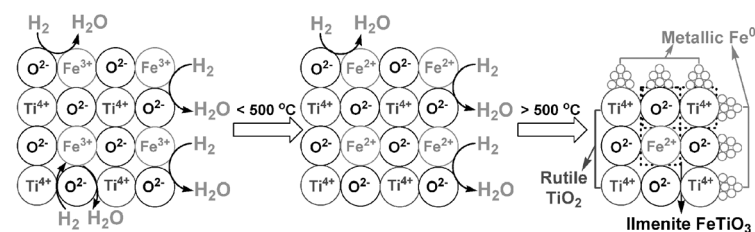
tallites in the FeTiO<sub>x</sub> catalyst results in the specific deoxidation/redox behavior of the Fe species, thus leading to high NH<sub>3</sub>-SCR performance at low temperatures.

## Conclusions

The advantageous combination of Fe species and Ti species in the FeTiO<sub>x</sub> catalyst resulted in its high NO<sub>x</sub> reduction efficiency in the NH<sub>3</sub> selective catalytic reduction (NH<sub>3</sub>-SCR) of NO<sub>x</sub> with a broad operation temperature window and good SO<sub>2</sub> durability. The unique edge-shared Fe<sup>3+</sup>–(O)<sub>2</sub>–Ti<sup>4+</sup> structure in iron titanate crystallites was confirmed to be the true catalytically active site, with an electronic inductive effect between Fe<sup>3+</sup> species and Ti<sup>4+</sup> species leading to the enhanced oxidation ability of Fe<sup>3+</sup> species. In the H<sub>2</sub>-TPR process, the Fe<sup>3+</sup> species in the FeTiO<sub>x</sub> catalyst can be totally reduced to Fe<sup>2+</sup> species below 500 °C in the form of ilmenite FeTiO<sub>3</sub>, whereas the Fe<sup>3+</sup> species in Fe<sub>2</sub>O<sub>3</sub> can only be reduced to Fe<sub>3</sub>O<sub>4</sub>. The typical NH<sub>3</sub>-SCR reaction mainly occurs below 500 °C, and the enhanced redox ability of Fe<sup>3+</sup> species in FeTiO<sub>x</sub> catalyst is bene-



**Figure 7.** A) Fourier transforms of filtered EXAFS oscillations  $k^3\chi(k)$  into  $R$  space and B) corresponding filtered  $k^3\chi(k)$  of TiK edge in Ti-containing samples in the  $k$  range of 2.5–13  $\text{\AA}^{-1}$ . The dotted lines correspond to the curve-fitting results in  $k$  space calculated by using the FEFF 8.4 code. ● Indicate rutile  $\text{TiO}_2$ ; ∇  $\text{FeTiO}_3$ .



**Scheme 1.** Proposed  $\text{H}_2$  reduction process of the  $\text{FeTiO}_x$  catalyst.

ficial to enhance its catalytic performance. Based on XANES linear-fitting and EXAFS curve-fitting results, the unique deoxidation behavior of the  $\text{FeTiO}_x$  catalyst in the  $\text{H}_2$ -TPR process could be clearly elucidated. The results in this study can supply some ideas for the investigation of the local structure and redox ability of active sites in mixed oxide catalysts simultaneously for certain catalytic reactions.

## Experimental Section

### Catalyst preparation, $\text{NH}_3$ -SCR performance test, and HRTEM characterization

$\text{FeTiO}_x$  catalyst with Fe/Ti molar ratio of 1:1 was prepared by a coprecipitation method using  $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$  and  $\text{Ti}(\text{SO}_4)_2$  as precursors and  $\text{NH}_3\text{-H}_2\text{O}$  (25 wt%) as precipitator. The resulting precipitate was filtrated, washed with distilled water, dried at  $100^\circ\text{C}$  for 12 h, and finally calcined in air at  $400^\circ\text{C}$  for 6 h. As reference samples, pristine hematite  $\text{Fe}_2\text{O}_3$  and anatase  $\text{TiO}_2$  were also prepared by the same method. The steady-state  $\text{NH}_3$ -SCR performance of prepared samples was tested in a fixed-bed quartz tube reactor at atmospheric pressure, and the reaction conditions of standard SCR

were controlled as follows:  $[\text{NO}] = [\text{NH}_3] = 500 \text{ ppm}$ ,  $[\text{O}_2] = 5 \text{ vol\%}$ , and  $\text{N}_2$  balance; 0.6 mL catalyst, 20–40 mesh; total flow rate of  $500 \text{ mL min}^{-1}$ , and gas hourly space velocity (GHSV) of  $50000 \text{ h}^{-1}$ . The effluent gas was continuously analyzed by using an FTIR spectrometer (Nicolet Nexus 670) equipped with a heated, low-volume multiple-path gas cell (2 m). The HRTEM images of the  $\text{FeTiO}_x$  catalyst, hematite  $\text{Fe}_2\text{O}_3$ , and anatase  $\text{TiO}_2$  with high magnification were obtained on a JEOL 2011 TEM (JEOL).

### $\text{H}_2$ -TPR experiments

Prior to the  $\text{H}_2$ -TPR procedure, the samples (100 mg) were pretreated in a flow of 20 vol%  $\text{O}_2/\text{He}$  ( $30 \text{ mL min}^{-1}$ ) at  $300^\circ\text{C}$  for 0.5 h and cooled down to RT. Then the temperature was raised linearly to  $900^\circ\text{C}$  at a rate of  $10^\circ\text{C min}^{-1}$  in a flow of 5 vol%  $\text{H}_2/\text{Ar}$  ( $30 \text{ mL min}^{-1}$ ). The  $\text{H}_2$  signal ( $m/z=2$ ) was monitored online by using a quadrupole mass spectrometer (HPR20, Hiden Analytical Ltd.) to obtain the  $\text{H}_2$ -TPR profiles. After the reduction temperature reaching  $900^\circ\text{C}$ , the reduced  $\text{Fe}_2\text{O}_3$  and  $\text{FeTiO}_x$  samples were naturally cooled to RT under the  $\text{H}_2$  protection, and then put into sealed bags for the following XAFS measurements.

### XAFS measurements

The XANES and EXAFS spectra of FeK and TiK edges were measured in transmission mode at RT on the BL-7C beam line, Photon Factory, KEK, Japan and BL14W1 beam line, Shanghai Synchrotron Radiation Facility (SSRF). The storage ring was operated at 2.5 GeV with 300 mA as an average storage current for the BL-7C beam line and 3.5 GeV with 200 mA as an average storage current for the BL14W1 beam line. The synchrotron radiation beam line was monochromatized with a Si (111) double crystal monochromator, and mirrors were used to eliminate higher harmonics. The incident and transmitted beam intensities were monitored using ionization chambers filled with pure  $\text{N}_2$ . XAFS data were analyzed by using the REX2000 program (Rigaku Co.). XANES spectra were normalized with edge height. EXAFS oscillation  $\chi(k)$  was extracted by using spline smoothing with a Cook–Sayers criterion<sup>[43]</sup> and weighted by  $k^3$  in order to compensate for the diminishing amplitude in the high  $k$  range because of the decay of the photoelectron wave. Thereafter, the filtered  $k^3$ -weighted  $\chi(k)$  was Fourier transformed into  $R$  space ( $k$  range: 2.5–15  $\text{\AA}^{-1}$  for FeK EXAFS and 2.5–13  $\text{\AA}^{-1}$  for TiK EXAFS) with a Hanning function window. In the curve-fitting step, the possible backscattering amplitude and phase shift were calculated by using the FEFF 8.4 code.<sup>[44]</sup> For standard reference samples, the coordination numbers (CN), the bond lengths ( $R$ ), the edge corrections ( $\Delta E$ ) and the Debye–Waller factors were all set to be adjustable. And for the catalysts containing confirmed components by XANES linear fitting results, the bond lengths of specific shells were fixed exactly to the



**Table 3.** Calibrated EXAFS curve-fitting results of TiK edge in different samples (M=Fe or Ti)

Sample	Ref. <sup>[a]</sup>	Shell	CN <sup>[b]</sup>	R <sup>[c]</sup> [Å]	ΔE [eV]	DW <sup>[d]</sup> [Å]	R factor [%]
Ti foil	Ti	Ti-Ti	6.0 ± 0.9	2.90 ± 0.01	4.4 ± 1.3	0.095 ± 0.015	1.2
anatase TiO <sub>2</sub>	anatase TiO <sub>2</sub>	Ti-O	6.0 ± 1.0	1.98 ± 0.01	12.5 ± 2.0	0.101 ± 0.019	1.7
		Ti-O-Ti <sub>1</sub>	3.0 ± 0.9	3.02 ± 0.02	4.4 ± 2.9	0.077 ± 0.027	
		Ti-O-Ti <sub>2</sub>	3.0 ± 1.6	4.03 ± 0.03	-7.6 ± 4.1	0.088 ± 0.043	
rutile TiO <sub>2</sub>	rutile TiO <sub>2</sub>	Ti-O	6.0 ± 1.3	1.98 ± 0.02	-6.4 ± 2.9	0.096 ± 0.024	2.9
		Ti-O-Ti <sub>1</sub>	2.0 ± 1.2	2.96 ± 0.04	-17.1 ± 6.5	0.071 ± 0.059	
		Ti-O-Ti <sub>2</sub>	4.0 ± 1.4	3.57 ± 0.02	-13.0 ± 3.2	0.066 ± 0.035	
FeTiO <sub>3</sub>	FeTiO <sub>3</sub>	Ti-O	6.0 ± 1.6	1.86 ± 0.02	-17.2 ± 4.3	0.082 ± 0.039	3.6
		Ti-O-M <sub>1</sub>	4.0 ± 0.8	2.96 ± 0.01	-8.4 ± 2.0	0.060 ± 0.024	
		Ti-O-M <sub>2</sub>	3.0 ± 1.3	3.38 ± 0.03	9.9 ± 3.5	0.081 ± 0.037	
Fe <sub>2</sub> TiO <sub>5</sub>	Fe <sub>2</sub> TiO <sub>5</sub>	Ti-O	6.0 ± 1.1	1.96 ± 0.02	-8.4 ± 2.7	0.105 ± 0.023	3.3
		Ti-O-M <sub>1</sub>	3.0 ± 1.4	3.17 ± 0.03	-8.1 ± 4.5	0.082 ± 0.036	
		Ti-O-M <sub>2</sub>	3.0 ± 2.8	3.72 ± 0.05	11.0 ± 7.2	0.071 ± 0.094	
FeTiO <sub>x</sub>	Fe <sub>2</sub> TiO <sub>5</sub>	Ti-O	6.9 ± 1.7	1.97 ± 0.02	-5.8 ± 2.5	0.125 ± 0.021	6.9
		Ti-O-Fe	2.0 ± 1.7	3.08 ± 0.04	-2.6 ± 2.3	0.119 ± 0.052	
FeTiO <sub>x</sub> -H <sub>2</sub>	FeTiO <sub>3</sub> *rutile TiO <sub>2</sub>	Ti-O	2.8 ± 1.7	1.86 ± 0.02	-13.0 ± 7.5	0.062 ± 0.065	5.8
		Ti-O-M <sub>1</sub>	1.9 ± 0.4	2.96 ± 0.01	-7.6 ± 1.6	0.043 ± 0.022	
		Ti-O-M <sub>2</sub>	1.5 ± 1.5	3.38 ± 0.03	0.0 ± 0.0	0.072 ± 0.049	
		*Ti-O	3.2 ± 0.7	1.98 ± 0.02	-7.5 ± 11.2	0.111 ± 0.036	
		*Ti-O-Ti <sub>1</sub>	1.1 ± 0.7	2.96 ± 0.04	-25.6 ± 15.2	0.095 ± 0.023	
		*Ti-O-Ti <sub>2</sub>	2.1 ± 0.7	3.57 ± 0.02	19.7 ± 1.0	0.102 ± 0.020	

[a] For information on the calibration of (reference) coordination numbers and the treatment of error margins, see the Experimental Section. [b] Coordination number. [c] Bond length. [d] Debye-Waller factor.

values derived from reference samples, and the rest parameters were set to be adjustable. Thus, we adopted the same error intervals of the fixed bond lengths for the studied catalysts as those in reference samples. During this curve-fitting step, the numbers of free parameters for all samples comply with the Nyquist criterion. After the EXAFS curve fitting of standard reference samples (including Fe foil, FeO, FeTiO<sub>3</sub>, Fe<sub>2</sub>O<sub>3</sub>, Fe<sub>2</sub>TiO<sub>5</sub>, Ti foil, anatase TiO<sub>2</sub>, and rutile TiO<sub>2</sub>), their coordination numbers and bond lengths were all calibrated to the corresponding values in theoretical crystal structures. Thus, more reasonable fitting data could be obtained that were closer to the real situations. Based on the calibrated fitting data of standard reference samples, both the coordination numbers and bond lengths as well as the corresponding error intervals in the studied catalysts were also calibrated.

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